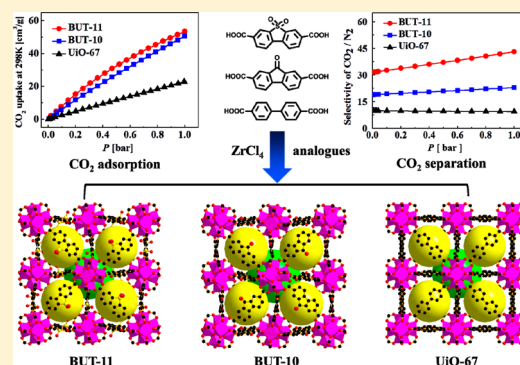


Tuning CO₂ Selective Adsorption over N₂ and CH₄ in UiO-67 Analogues through Ligand FunctionalizationBin Wang,^{†,‡} Hongliang Huang,^{†,§} Xiu-Liang Lv,[†] Yabo Xie,[†] Ming Li,^{*,‡} and Jian-Rong Li^{*,†,§}[†]Beijing Key Laboratory for Green Catalysis and Separation and Department of Chemistry and Chemical Engineering, Beijing University of Technology, Beijing 100124, P. R. China[‡]College of Chemistry and Molecular Engineering, Qing Dao University of Science and Technology, Qingdao 266042, P. R. China[§]State Key Laboratory of Organic–Inorganic Composites, Beijing University of Chemical Technology, Beijing 100029, P. R. China

Supporting Information

ABSTRACT: Introducing functional groups into pores of metal–organic frameworks (MOFs) through ligand modification provides an efficacious approach for tuning gas adsorption and separation performances of this type of novel porous material. In this work, two UiO-67 analogues, [Zr₆O₄(OH)₄(FDCA)₆] (BUT-10) and [Zr₆O₄(OH)₄(DTDAO)₆] (BUT-11), with functionalized pore surfaces and high stability were synthesized from two functional ligands, 9-fluorenone-2,7-dicarboxylic acid (H₂FDCA) and dibenzo[*b,d*]thiophene-3,7-dicarboxylic acid 5,5-dioxide (H₂DTDAO), respectively, and structurally determined by single-crystal X-ray diffraction. Notwithstanding skeleton bend of the two ligands relative to the linear 4,4'-biphenyldicarboxylic acid in UiO-67, the two MOFs have structures similar to that of UiO-67, with only lowered symmetry in their frameworks. Attributed to these additional functional groups (carbonyl and sulfone, respectively) in the ligands, BUT-10 and -11 show enhanced CO₂ adsorption and separation selectivities over N₂ and CH₄, in spite of decreased pore sizes and surface areas compared with UiO-67. At 298 K and 1 atm, the CO₂ uptake is 22.9, 50.6, and 53.5 cm³/g, and the infinite dilution selectivities of CO₂/CH₄ are 2.7, 5.1, and 9.0 and those of CO₂/N₂ are 9.4, 18.6, and 31.5 for UiO-67, BUT-10, and BUT-11, respectively. The selectivities of CO₂/CH₄ and CO₂/N₂ are thus enhanced 1.9 and 2.0 times in BUT-10 and 3.3 and 3.4 times in BUT-11, respectively, on the basis of UiO-67. The adsorption mechanism of CO₂ in BUT-11 has also been explored through computational simulations. The results show that CO₂ molecules locate around the sulfone groups in pore surfaces of BUT-11, verifying at the molecular level that sulfone groups significantly increase the affinity toward CO₂ molecules of the framework. This provides thus an efficient strategy for the design of CO₂ capture materials.



1. INTRODUCTION

Concerns about greenhouse gases in the atmosphere have led to the development of efficient CO₂ capture technologies.^{1,2} For the separation of CO₂, adsorption-based methods are promising because of their simple and easy control, low operating and capital costs, and superior energy efficiency.³ Among the various materials for adsorptive separation, metal–organic frameworks (MOFs), as a new family of porous solids, are attractive because of their large surface areas, adjustable pore sizes, and controllable pore-surface properties. So far, a lot of papers including several excellent reviews have demonstrated the potential application of MOFs in CO₂ capture and storage.^{4–8} Nevertheless, one drawback faced by most MOFs for application in CO₂ capture lies in their poor physicochemical stability.⁹

Recently, a zirconium(IV)-based MOF named UiO-66 has attracted intense interest for its excellent stability and good adsorption ability toward some small molecules including CO₂.^{10–12} It has been demonstrated that this MOF is thermally stable up to 773 K and its structure remains unaltered toward a wide class of aqueous solvents, in contrast to the majority of

MOFs. The hydroxylated form of UiO-66 is constructed from hexanuclear [Zr₆O₄(OH)₄] building units, in which the triangular faces of the Zr₆ octahedron are alternatively capped by μ₃-O and μ₃-OH groups. The Zr₆ polyhedra are interconnected along the edges through carboxylates of 12 1,4-benzenedicarboxylate (BDC) linkers to form a cubic three-dimensional (3D) framework.¹⁰ Some research has shown that the introduction of functional groups such as –COOH, –SO₃H, and –NH₂ to the BDC ligand could significantly enhance the CO₂ adsorption and separation ability of UiO-66 in either experiments or theoretical calculations.^{13–16} For gas adsorptive separation, both the adsorption capacity and selectivity play important roles. Because of smaller pore size and surface area, the introduction of functional groups can significantly reduce the specific surface area of UiO-66, which would lead to a negative effect on its adsorption working capacity.^{12,13} UiO-67, the isostructural analogue of UiO-66, is constructed from a longer linker, 4,4'-biphenyldicarboxylate

Received: June 9, 2014

Published: August 12, 2014

(BPDC), to give larger pore size, being considered as a more promising platform for functional materials with enhanced CO₂ adsorption and separation abilities.¹⁷ However, owing to the larger pore size in UiO-67, the interaction between the CO₂ molecule and the framework is weaker than that in UiO-66. Thus, loading polar functional groups on the BPDC skeleton is a feasible method to increase the adsorption uptake and selectivity of the resulting material toward CO₂. To the best of our knowledge, there are rare reports on the functionalized UiO-67 analogues.¹⁸

In this work, by employing two functionalized ligands, 9-fluorenone-2,7-dicarboxylic acid (H₂FDCA) and dibenzo[*b,d*]-thiophene-3,7-dicarboxylic acid 5,5-dioxide (H₂DTDAO), two UiO-67 analogues, [Zr₆O₄(OH)₄(FDCA)₆]_{*n*}·S (BUT-10, where BUT = Beijing University of Technology) and [Zr₆O₄(OH)₄(DTDAO)₆]_{*n*}·S (BUT-11), were synthesized, respectively. Their structures were determined by single-crystal X-ray diffraction. Compared with UiO-67, BUT-10 and -11 show large enhancement of the CO₂ adsorption and selectivity over CH₄ and N₂. It should be pointed out that, during the preparation of this manuscript, CO₂ adsorption of BUT-11 was reported,¹⁸ where powder X-ray diffraction (PXRD) was used to determine its structure and CO₂ adsorption data were provided. Herein, we report the synthesis and crystal structures of BUT-10 and -11, which were characterized by single-crystal X-ray diffraction, as well as a detailed CO₂ adsorption and separation exploration by both experimental characterizations and computational simulations.

2. EXPERIMENTAL SECTION

2.1. Materials and Methods. All general reagents and solvents (AR grade) were commercially available and were used as received. ZrCl₄, 4,4'-biphenyldicarboxylic acid (H₂BPDC), and 9-fluorenone-2,7-dicarboxylic acid (H₂FDCA) were purchased from J&K Chemical. Fourier transform infrared (FT-IR) data were recorded on an IRAffinity-1 instrument. PXRD patterns were recorded on a Bruker D8-Focus Bragg–Brentano X-ray powder diffractometer equipped with a copper sealed tube ($\lambda = 1.54178$) at room temperature. Simulation of the PXRD spectrum was carried out by the single-crystal data and diffraction-crystal module of the *Mercury* program available free of charge via the Internet at <http://www.ccdc.cam.ac.uk/mercury/>. Thermogravimetric analysis (TGA) data were obtained on a TGA-50 (Shimadzu) thermogravimetric analyzer with a heating rate of 2 °C/min under a N₂ atmosphere. Gas adsorption isotherms were measured by a volumetric method using a Micromeritics ASAP2020 surface area and pore analyzer. A high purity grade of gases (99.999% purity) was used throughout the adsorption experiments. *P*₀ is the saturation pressure at given temperature of the respective gas. For activation of the samples, the as-synthesized sample was soaked in fresh *N,N*-dimethylformamide (DMF) for 24 h, and the extract was discarded. Fresh acetone was subsequently added, and the sample was allowed to remain in it for 24 h. This procedure was again repeated two times. After the acetone extract was decanted, the sample was dried under a dynamic vacuum (<10⁻³ Torr) at room temperature for 1 h. Before adsorption measurement, the sample was further activated using the “outgas” function of the adsorption analyzer for 10 h at 80 °C.

2.2. Syntheses. The ligand dibenzo[*b,d*]thiophene-3,7-dicarboxylic acid 5,5-dioxide (H₂DTDAO)¹⁹ and UiO-67²⁰ were synthesized according to the literature methods.

[Zr₆O₄(OH)₄(FDCA)₆]_{*n*}·S (BUT-10; S = Nonassignable Solvent Molecules). A solution of H₂FDCA (80 mg, 0.3 mmol), ZrCl₄ (71 mg, 0.3 mmol), and 3 mL of acetic acid (HOAc) in 17 mL of DMF was sealed in a 20 mL glass vial and heated at 120 °C for 10 h. The resulting yellow crystals were collected and washed with DMF and acetone and then dried in air (yield: 49 mg). For PXRD of the as-

synthesized material, TGA, and FT-IR, see Figures S1–S3 in the experimental section of the Supporting Information (SI), respectively.

[Zr₆O₄(OH)₄(DTDAO)₆]_{*n*}·S (BUT-11). A solution of H₂DTDAO (61 mg, 0.2 mmol), ZrCl₄ (47 mg, 0.2 mmol), and 1.7 mL of trifluoroacetic acid in 18 mL of DMF was sealed in a 20 mL glass vial and heated at 120 °C for 48 h. The resulting colorless crystals were collected and washed with DMF and acetone and then dried in air (yield: 35 mg). For PXRD of as-synthesized material, TGA, and FT-IR, see Figures S1–S3 in the experimental section of the SI, respectively.

2.3. Single-Crystal X-ray Diffraction. The diffraction data of BUT-10 were collected on a Bruker-AXS APEX-II CCD X-ray diffractometer equipped with graphite-monochromated Mo *K* α radiation ($\lambda = 0.71073$ Å) at 173 K. Raw data collection and reduction were done using *APEX2* software.²¹ Absorption corrections were applied using the *SADABS* routine.²² The diffraction data of BUT-11 were collected on an Agilent Supernova CCD diffractometer equipped with graphite-monochromated Cu *K* α radiation ($\lambda = 1.54184$ Å) at 173 K. The data sets were corrected by empirical absorption correction using spherical harmonics, implemented in the *SCALE3 ABSPACK* scaling algorithm.²³ The structures of BUT-10 and -11 were solved by direct methods and refined by full-matrix least squares on *F*² with anisotropic displacement using the *SHELXTL* software package.²⁴ Non-H atoms were refined with anisotropic displacement parameters during the final cycles. H atoms of ligands were calculated in ideal positions with isotropic displacement parameters. Those in the OH groups of the zirconium(IV)-based clusters were not added but were calculated into the molecular formula of the crystal data. There are large solvent-accessible pore volumes in the crystals of BUT-10 and -11, which are occupied by highly disordered solvent molecules. No satisfactory disorder model for these solvent molecules could be achieved, and therefore the *SQUEEZE* program implemented in *PLATON* was used to remove the electron densities of these disordered species.²⁵ Thus, all of the electron densities from free solvent molecules have been “squeezed” out. The details of structural refinement can be found in Tables S1 and S2 and the CIF files in the SI.

2.4. Fitting of Pure-Component Isotherms. The pure-component CO₂, CH₄, and N₂ adsorption isotherm data measured at 298 K were fitted with the dual-site Langmuir model

$$N = N_1^{\max} \frac{b_1 p}{1 + b_1 p} + N_2^{\max} \frac{b_2 p}{1 + b_2 p}$$

where *p* is the pressure of the bulk gas at equilibrium with the adsorbed phase (kPa), *N* is the adsorbed amount per mass of adsorbent (mol/kg), *N*₁^{max} and *N*₂^{max} are the saturation capacities of sites 1 and 2 (mol/kg), and *b*₁ and *b*₂ are the affinity coefficients of sites 1 and 2 (kPa⁻¹).

2.5. Isotheric Heat of Adsorption. The isotheric heat of adsorption represents the strength of the interactions between adsorbate molecules and the adsorbent lattice atoms and can be used as a measurement of the energetic heterogeneity of a solid surface. The isotheric heat of adsorption at a given amount can be calculated by the Clausius–Clapeyron equation as

$$Q_{\text{st}} = -RT^2 \left(\frac{\partial \ln P}{\partial T} \right)_{n_a}$$

where *Q*_{st} is the isotheric heat of adsorption (kJ/mol), *P* is the pressure (kPa), *T* is the temperature, *R* is the gas constant, and *n*_a is the adsorption amount (mmol/g).

2.6. Calculations of the Adsorption Selectivity. The selectivity of preferential adsorption of component 1 over component 2 in a mixture containing 1 and 2 can be formally defined as

$$S_{\text{ads}} = \frac{q_1/q_2}{p_1/p_2}$$

In the equation, *q*₁ and *q*₂ are the absolute loadings. In all of the calculations to be presented below, the calculations of *S*_{ads} are based on

the use of the Ideal Adsorbed Solution Theory (IAST) of Myers and Prausnitz.²⁶ These calculations are carried out using the pure-component isotherm fits of absolute loadings.

3. RESULTS AND DISCUSSION

3.1. Synthesis, Structures, and Pore Characterizations.

The optimized geometries of the three ligands used are shown in Figure 1a. It can be seen that a carbonyl group and a sulfone

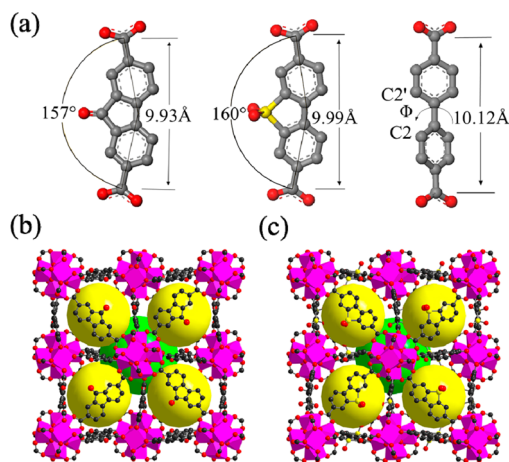


Figure 1. (a) Geometries of the ligands FDCA, DTDAO, and BPDC. (b) Crystal structure of BUT-10. (c) Crystal structure of BUT-11. H atoms on the ligands are omitted for clarity; the large green and yellow spheres represent void regions inside the frameworks. Color code: Zr polyhedra, pink; octahedral cages, green; tetrahedral cages, yellow; C, black gray; O, red; S, yellow.

group are introduced into the BPDC skeleton in FDCA and DTDAO, respectively. These introduced functional groups link C2 and C2' in BPDC to make the skeleton bent with an angle of 157° and 160°, respectively, and the distance between the two carboxylate groups become slightly shorter compared with that in BPDC. In addition, two benzene rings in FDCA and DTDAO are coplanar, in contrast with an about 31° dihedral angle in BPDC. It is interesting that, even with changes of these structural parameters in the ligands, the framework structures of their Zr-MOFs are similar, as described below. This provides a good platform for exploring the effects of the functional groups in ligands on the gas adsorption performances of the resulting MOFs.

The reaction of the two functional ligands with $ZrCl_4$ in DMF at 120 °C yielded two MOFs, BUT-10 and -11, as single crystals, respectively. It should be pointed out that, during the synthesis of the zirconium-based MOFs, a modular acid is crucial, particularly for obtaining a single-crystal sample. In our case, acetic acid and trifluoroacetic acid were used, respectively, to obtain MOF crystals big enough for single-crystal X-ray diffraction. It has been reported that BUT-11 can also be obtained in microcrystal form when using HCl or AcOH as a modular.¹⁸ In the case when HCl was used, it was found that Cl^- ions could provide charge compensation to the Zr-MOF structures, which makes the structure break down incompletely.²⁷

As shown in Figure S2 in the SI, BUT-10 and -11 are thermally stable up to 500 and 420 °C, respectively, being slightly lower than that of UiO-67 (decomposition at 520 °C). Clearly, the functionalization of the ligand led to a change of the coordination bond strength, which finally affected the

materials' thermal stability. After removal of the solvent molecules from the as-synthesized samples, the frameworks of BUT-10 and -11 remained intact, as confirmed by PXRD (Figure S1 in the SI).

Single-crystal X-ray diffraction analysis reveals that the structures of BUT-10 and -11 were constructed by hexanuclear $[Zr_6O_4(OH)_4]$ building units interconnected through carboxylate groups of 12 organic linkers to form 3D frameworks (Figure 1b,c). Obviously, the two MOFs share the same structure with UiO-67 even though of lower symmetry [crystallized in the lower space group of $Pa\bar{3}$ compared with that ($Fm\bar{3}m$) of UiO-67] because of the geometric change of the ligands. This result proves that the structures of MOFs can be tolerant in a certain degree change of ligands in some systems. In their frameworks, there exist two types of polyhedral cages, octahedral and tetrahedral (Figures S5 and S6 in the SI). Their sizes across respective edges are 19.16, 18.42, and 18.54 Å for UiO-67, BUT-10, and BUT-11, respectively. After removal of free solvent molecules, the total solvent-accessible volumes of these three frameworks are estimated to be 68.1, 64.0, and 63.4%, respectively, by PLATON.

In order to evaluate the permanent porosity of the two MOFs, N_2 adsorption experiments were performed at 77 K. For comparison, N_2 adsorption of UiO-67 was also measured under similar conditions. As shown in Figure 2, the three MOFs gave

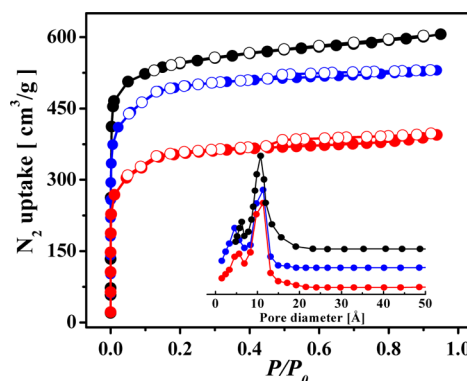


Figure 2. N_2 sorption isotherms of UiO-67 (black), BUT-10 (blue), and BUT-11 (red) at 77 K (filled and open symbols represent adsorption and desorption data, respectively). Inset: density functional theory pore-size distributions for the three MOFs determined from their N_2 adsorption isotherms at 77 K.

basically type I isotherms, being characteristic of microporous materials. Small stepwise adsorption and hysteresis observed in BUT-10 and -11 could be attributed to cage-shape pores in their frameworks.²⁸ The evaluated Brunauer–Emmett–Teller (S_{BET}) and Langmuir surface areas ($S_{Langmuir}$) from the N_2 adsorption isotherms are 2505 and 3040, 1848 and 2185, and 1310 and 1571 m^2/g for UiO-67, BUT-10, and BUT-11, respectively. The maximum N_2 adsorptions (at $P/P_0 = 0.95$) of 606, 531, and 398 cm^3/g suggest pore volumes of 1.10, 0.82, and 0.62 cm^3/g for UiO-67, BUT-10, and BUT-11, respectively. It is clear that the introduction of functional groups in the ligands decreased the pore sizes and surface areas of the resulting MOFs. The pore-size distributions are also consistent with the crystal structure determination (Figure 2, inset).

3.2. CO_2 Selective Adsorption. The adsorption isotherms of pure CO_2 , CH_4 , and N_2 in the evacuated UiO-67, BUT-10, and BUT-11, respectively, were experimentally measured, and

the results are shown in Figures 3 (at 298 K) and S4 in the SI (at 273 K). As can be seen from these isotherm data, the three

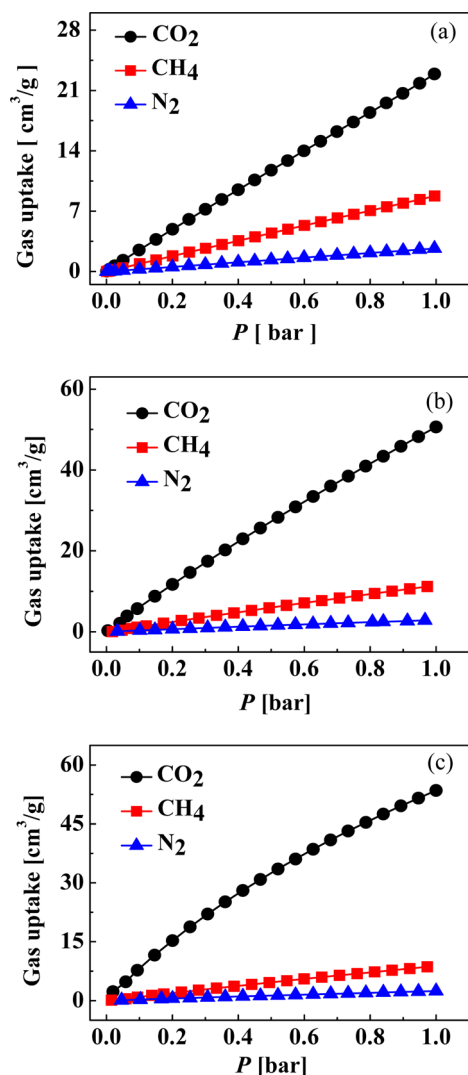


Figure 3. CO₂, CH₄, and N₂ adsorption isotherms at 298 K for (a) UiO-67, (b) BUT-10, and (c) BUT-11.

MOFs have a maximum CO₂ uptake of 22.9, 50.6, and 53.5 cm³/g at 298 K and 1 atm, CH₄ uptake of 8.7, 11.4, and 8.8 cm³/g, and N₂ uptake of 2.7, 2.9, and 2.4 cm³/g. It is quite interesting that the CO₂ uptakes in BUT-10 and -11 are now more than double compared with their parent MOF, UiO-67, demonstrating that added functional groups in the ligands are able to significantly enhance the CO₂ adsorption capacities of the resulting MOFs, even if they have led to decreased pore sizes as mentioned above. In a recent report,¹⁸ the CO₂ uptake at 298 K and 1 atm was 53.1 cm³/g for the BUT-11 sample obtained using HCl as the modular in the synthesis, being comparable with our result.

The isosteric heats of CO₂ adsorption are calculated based on the adsorption data collected at 273 and 298 K. As shown in Figure 4, the Q_{st} values for BUT-10 and -11 are 21.8–18.9 and 25.9–22.3 kJ/mol, respectively, being higher than that of 15.9–15.8 kJ/mol for UiO-67. The trend in the Q_{st} values is clearly in accordance with the uptake amounts of the order: BUT-11 > BUT-10 > UiO-67.

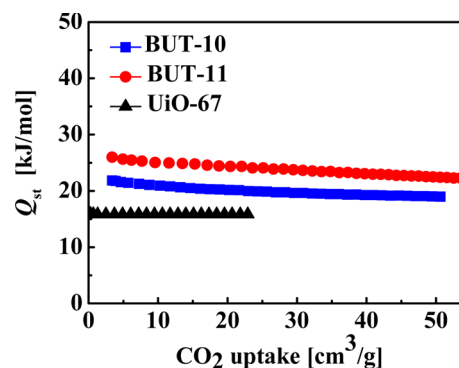


Figure 4. Isosteric heats (Q_{st}) of CO₂ adsorption in UiO-67, BUT-10, and BUT-11.

It is clear that the CO₂ uptake and isosteric heats of BUT-10 and -11 are higher than those of UiO-67, which can be attributed to the stronger interactions between the carbonyl and sulfone groups in the two MOFs and CO₂ molecules. In order to elucidate these observations at the molecular level, grand canonical Monte Carlo (GCMC) simulation was used to evaluate the adsorption properties of BUT-11.²⁹ As shown in Figure 5, after adsorption snapshots indicate that CO₂

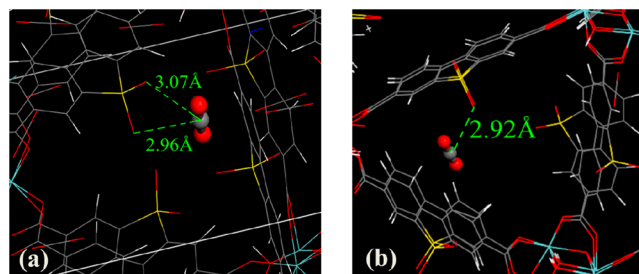


Figure 5. Local view of the snapshots extracted from GCMC simulation at 0.1 bar and 298 K, showing the interactions between the CO₂ molecule and the sulfone group in BUT-11: (a) cross interaction; (b) parallel interaction.

molecules locate around sulfone groups, which shows that there are strong interactions between the CO₂ molecule and the sulfone group in BUT-11. We can also see that there are two types of interaction modes: cross-interaction mode (Figure 5a) and parallel-interaction mode (Figure 5b) for CO₂. In the former mode, the linear O=C=O molecule displays a cross interaction with the O=S=O entity, with the distance between the C atom of CO₂ and the two O atoms of the sulfone group being 2.96 and 3.07 Å, respectively. In the latter mode, it displays a parallel interaction with the O=S=O entity, with a distance between the C and O atoms of 2.92 Å. The relatively short distance between the sulfone group and CO₂ molecule implies a strong interaction. Because the CO₂ molecule possesses a large quadrupole moment, the C atom is positively charged, and the O atoms of the sulfone group is negative, the interactions between the O atoms in sulfone and the C atom in CO₂ thus lead to two types of interaction modes. In addition, the radial distribution functions (RDFs) gave the distribution of CO₂ molecules around the sulfone group from the statistics perspective (see Figure S8 in the SI). It was found that the distance between the O atoms in the sulfone group and the C atom in CO₂ is about 3 Å, being in accordance with the snapshot results.

The enhancement of CO₂ uptakes found in the two functionalized MOFs prompted us to investigate their CO₂ adsorption selectivity over other important gases, including CH₄ and N₂. In order to establish the feasibility of these separations, we performed selectivity calculations using the IAST method.^{30–33} To obtain excellent fitting curves, several isotherm models were tested to fit the experimental pure isotherms of CO₂, CH₄, and N₂. For all MOFs, the dual-site Langmuir model was found to be best ($R^2 > 0.9995$; Figure S9 in the SI). The fitted isotherm parameters were used to predict the IAST selectivities of these MOFs. The resulting selectivities for 10:90 CO₂/CH₄ and 15:85 CO₂/N₂ mixtures in the three MOFs as a function of the total bulk pressure are presented in Figure 6. The CO₂/CH₄ selectivities are 2.7–2.9, 5.1–5.2, and

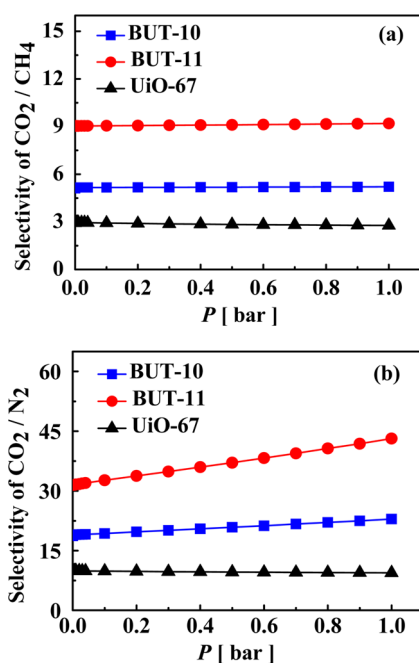


Figure 6. IAST-predicted selectivities toward (a) 10:90 CO₂/CH₄ and (b) 15:85 CO₂/N₂ in UiO-67, BUT-10, and BUT-11, respectively.

9.0–9.2, and the CO₂/N₂ selectivities are 9.4–9.9, 18.6–22.9, and 31.5–43.1 for UiO-67, BUT-10, and BUT-11, respectively. At very low loading, these values are almost identical with the ratios of the Henry constants obtained in the same ranges: 3.1, 6.2, and 9.5 for the CO₂/CH₄ and 9.4, 18.1, and 30.5 for the CO₂/N₂ in UiO-67, BUT-10, and BUT-11, respectively.

The higher CO₂ adsorption enthalpies and selectivities over CH₄ and N₂ in BUT-10 and -11 compared with those of UiO-67 can be attributed to the stronger interactions between the functionalized frameworks and CO₂ molecules. For BUT-11, the primary interaction between CO₂ molecules and the framework should be the electron donor–acceptor effect, in which the O atoms of the sulfone group serve as the electron-donor center and the C atoms of CO₂ as the electron-acceptor center. In addition, the quadrupolar interaction between the sulfone group and CO₂ molecule should also contribute to the observed trend. Accordingly, a similar but weaker electron donor–acceptor interaction may be expected in the CO₂ adsorption by BUT-10 because of the lower electron-donating ability of the carbonyl group compared with the sulfone group.

A lot of MOFs have been checked for the selective adsorption of CO₂ over CH₄ and N₂, and Table 1 lists infinite

Table 1. Infinite Dilution Selectivities of CO₂/CH₄ and CO₂/N₂ in Some Selected MOFs

MOF	T/K	$S_{\text{CO}_2/\text{CH}_4}$	$S_{\text{CO}_2/\text{N}_2}$	ref
BUT-10	298	5.1	18.6	this work
BUT-11	298	9.0	31.5	this work
UiO-67	298	2.7	9.4	this work
BUT-11(AcOH)	298	6.9	24.1	18
UiO-66(Zr)	298	3.2	13.4	13
UiO-66(Zr)-Br	298	4.7	25.1	13
UiO-66(Zr)-NO ₂	298	5.1	26.4	13
UiO-66(Zr)-NH ₂	303	9.0	12	12
UiO-66(Zr)-(OH) ₂	303	9.0	12	12
UiO-66(Zr)-(CF ₃) ₂	303	7.5	12	12
UiO-66(Zr)-SO ₃ H	303	9.1	12	12
ZIF-8	298	2.6	8.4	34
ZIF-68	298	5.0	18.0	35
ZIF-69	298	5.2	20.0	35
ZIF-70	298	5.1	17.0	35
ZIF-79	298	6.2	23.0	35
ZIF-81	298	6.3	24.0	35
ZIF-100	298	5.9	25.0	36
ZIF-95	298	4.3	18.0	36
Cu-BTC	298	6.4		37
MOF-5	298	1.8		37
MOF-74(Co)	298		25.0	38
Cu-BTTriex	298		21.0	39
MOF-508b	303	3.0	3.8	40
MIL-101(Cr)	288	6.5	12.6	41
MIL-125-NH ₂	298		27.0	42
MIL-68-Al	303	9.0	37.0	43
MIL-47(V)	298	2.0		44

dilution selectivities of CO₂/CH₄ and CO₂/N₂ in BUT-10 and -11 and some selected MOFs. The selectivities in BUT-11 are close to the MOFs with –NH₂, –OH, and –SO₃H functional groups including UiO-66-X¹² and MIL-125-NH₂,⁴² whereas the selectivities in BUT-10 are similar to –Br- and –NO₂-functionalized MOFs such as UiO-66-Br and –NO₂¹³ and some ZIFs.^{34,35} These results show that both carbonyl and sulfone groups are indeed useful in tailoring MOFs for CO₂ capture and other specific adsorption-based applications.

4. CONCLUSIONS

In summary, two MOFs, BUT-10 and -11, were obtained by using carbonyl- and sulfone-functionalized ligands, respectively, and structurally characterized by single-crystal X-ray diffraction. They have the same structures as UiO-67 even with a lower framework symmetry due to ligand modifications. The introduction of these functional groups has decreased the pore sizes and surface areas of the resulting MOFs but enhanced CO₂ adsorption capacities and selectivities over CH₄ and N₂, compared with UiO-67. In addition, computational simulations were performed to provide a microcosmic mechanism of CO₂ adsorption in BUT-11. Both snapshots and RDFs showed at the molecular level that CO₂ molecules locate around sulfone groups in BUT-11, and there are strong interactions between them. The adsorption selectivities found in BUT-10 and -11 are also comparable to those reported MOFs with functional groups of –NH₂, –OH, –Br, –NO₂, and –SO₃H. The comparable or relatively high CO₂ adsorption selectivities in BUT-10 and -11 demonstrate that introducing carbonyl and sulfone groups onto the pore surfaces of MOFs

can significantly increase their affinity toward CO₂ molecules, thereby enhancing the CO₂ capture ability, which provided a strategy for the development of CO₂ capture materials.

■ ASSOCIATED CONTENT

■ Supporting Information

X-ray crystallographic data in CIF format, details of structure refinement, PXRD, TGA, FT-IR, gas adsorption, additional structural figures, molecular simulation methods, force field details, GCMC simulation, RDFs, and fitting adsorption data. This material is available free of charge via the Internet at <http://pubs.acs.org>.

■ AUTHOR INFORMATION

Corresponding Authors

*E-mail: liming928@qust.edu.cn.

*E-mail: jrlj@bjut.edu.cn.

Notes

The authors declare no competing financial interest.

■ ACKNOWLEDGMENTS

We are thankful for financial support from the Natural Science Foundation of China (Grants 21271015 and 21322601), the Program for New Century Excellent Talents in University (Grant NCET-13-0647), and the Beijing Municipal Natural Science Foundation (Grant 2132013).

■ REFERENCES

- (1) Yu, K. M. K.; Curcic, I.; Gabriel, J.; Tsang, S. C. E. *ChemSusChem* **2008**, *1*, 893–899.
- (2) Figueroa, J. D.; Fout, T.; Plasynski, S.; McIlvried, H.; Srivastava, R. D. *Int. J. Greenhouse Gas Control* **2008**, *2*, 9–20.
- (3) Choi, S.; Drese, J. H.; Jones, C. W. *ChemSusChem* **2009**, *2*, 796–854.
- (4) Li, J.-R.; Kuppler, R. J.; Zhou, H.-C. *Chem. Soc. Rev.* **2009**, *38*, 1477–1504.
- (5) Düren, T.; Bae, Y.-S.; Snurr, R. Q. *Chem. Soc. Rev.* **2009**, *38*, 1237–1247.
- (6) Li, J.-R.; Ma, Y.-G.; McCarthy, M. C.; Sculley, J.; Yu, J.; Jeong, H.-K.; Balbuena, P. B.; Zhou, H.-C. *Coord. Chem. Rev.* **2011**, *255*, 1791–1823.
- (7) Li, J.-R.; Sculley, J.; Zhou, H.-C. *Chem. Rev.* **2012**, *112*, 869–932.
- (8) Sumida, K.; Rogow, D. L.; Mason, J. A.; McDonald, T. M.; Bloch, E. D.; Herm, Z. R.; Bae, T.; Long, J. R. *Chem. Rev.* **2012**, *112*, 724–781.
- (9) Czaja, A. U.; Trukhan, N.; Müller, U. *Chem. Soc. Rev.* **2009**, *38*, 1284–1293.
- (10) Cavka, J. H.; Jakobsen, S.; Olsbye, U.; Guillou, N.; Lamberti, C.; Bordiga, S.; Lillerud, K. P. *J. Am. Chem. Soc.* **2008**, *130*, 13850–13851.
- (11) Valenzano, L.; Civalieri, B.; Chavan, S.; Bordiga, S.; Nilsen, H. M.; Jakobsen, S.; Lillerud, K. P.; Lamberti, C. *Chem. Mater.* **2011**, *23*, 1700–1718.
- (12) Yang, Q.; Wiersum, A. D.; Llewellyn, P. L.; Guillerm, V.; Serre, C.; Maurin, G. *Chem. Commun.* **2011**, *47*, 9603–9605.
- (13) Zhang, W.; Huang, H.; Zhong, C.; Liu, D. *Phys. Chem. Chem. Phys.* **2012**, *14*, 2317–2325.
- (14) Kim, M.; Cohen, S. M. *CrystEngComm* **2012**, *14*, 4096–4104.
- (15) Kandiah, M.; Nilsen, M. H.; Usseglio, S.; Jakobsen, S.; Olsbye, U.; Tilset, M.; Larabi, C.; Quadrelli, E. A.; Bonino, F.; Lillerud, K. P. *Chem. Mater.* **2010**, *22*, 6632–6640.
- (16) Foo, M. L.; Horike, S.; Fukushima, T.; Hijikata, Y.; Kubota, Y.; Takata, M.; Kitagawa, S. *Dalton Trans.* **2012**, *41*, 13791–13794.
- (17) Yang, Q.; Guillerm, V.; Ragon, F.; Wiersum, A. D.; Llewellyn, P. L.; Zhong, C.; Devic, T.; Serre, C.; Maurin, G. *Chem. Commun.* **2012**, *48*, 9831–9833.
- (18) Xydias, P.; Spanopoulos, I.; Klontzas, E.; Froudakis, G. E.; Trikalitis, P. N. *Inorg. Chem.* **2014**, *53*, 679–681.
- (19) Neofotistou, E.; Malliakas, C. D.; Trikalitis, P. N. *Chem.—Eur. J.* **2009**, *15*, 4523–4527.
- (20) Schaate, A.; Roy, P.; Godt, A.; Lippke, J.; Waltz, F.; Wiebcke, M.; Behrens, P. *Chem.—Eur. J.* **2011**, *17*, 6643–6651.
- (21) APEX2 software package. Bruker Molecular Analysis Research Tool, version 2008.4; Bruker AXS Inc.: Madison, WI, 2008.
- (22) Sheldrick, G. M. SADABS, Program for Absorption Correction of Area Detector Frames; Bruker AXS Inc.: Madison, WI, 2001.
- (23) SCALE3 ABSPACK—An Oxford Diffraction program (1.0.4, gui:1.0.3) (C); Oxford Diffraction Ltd.: Oxford, U.K., 2005.
- (24) Sheldrick, G. M. SHELXTL 2008/4, Structure Determination Software Suite; Bruker AXS Inc.: Madison, WI, 2008.
- (25) Spek, A. L. *J. Appl. Crystallogr.* **2003**, *36*, 7–13.
- (26) Myers, A. L.; Prausnitz, J. M. *AIChE. J.* **1965**, *11*, 121–127.
- (27) DeCoste, J. B.; Peterson, G. W.; Jasuja, H.; Glover, T. G.; Huang, Y.; Walton, K. S. *J. Mater. Chem. A* **2013**, *1*, 5642–5650.
- (28) Katz, M. J.; Brown, Z. J.; Colón, Y. J.; Siu, P. W.; Scheidt, K. A.; Snurr, R. Q.; Hupp, J. T.; Farha, O. K. *Chem. Commun.* **2013**, *49*, 9449–9451.
- (29) The details of the computational strategies are provided in the SI.
- (30) Cavenati, S.; Grande, C. A.; Rodrigues, A. E. *J. Chem. Eng. Data* **2004**, *49*, 1095–1101.
- (31) Belmabkhout, Y.; Pirngruber, G.; Jolimaite, E.; Methivier, A. *Adsorption* **2007**, *13*, 341–349.
- (32) Krishna, R.; van Baten, J. M. *Phys. Chem. Chem. Phys.* **2011**, *13*, 10593–10616.
- (33) He, Y.; Xiang, S.; Zhang, Z.; Xiong, S.; Wu, C.; Zhou, W.; Yildirim, T.; Krishna, R.; Chen, B. *J. Mater. Chem. A* **2013**, *1*, 2543–2551.
- (34) Huang, H.; Zhang, W.; Liu, D.; Liu, B.; Chen, G.; Zhong, C. *Chem. Eng. Sci.* **2011**, *66*, 6297–6305.
- (35) Banerjee, R.; Furukawa, H.; Britt, D.; Knobler, C.; O’Keeffe, M.; Yaghi, O. M. *J. Am. Chem. Soc.* **2009**, *131*, 3875–3877.
- (36) Wang, B.; Côté, A. P.; Furukawa, H.; O’Keeffe, M.; Yaghi, O. M. *Nature* **2008**, *453*, 207–211.
- (37) Yang, Q.; Zhong, C. *J. Phys. Chem. B* **2006**, *110*, 17776–17783.
- (38) Cho, H.-Y.; Yang, D.-A.; Kim, J.; Jeong, S.-Y.; Ahn, W.-S. *Catal. Today* **2012**, *185*, 35–40.
- (39) Demessence, A.; D’Alessandro, M. D.; Foo, M. L.; Long, J. R. *J. Am. Chem. Soc.* **2009**, *131*, 8784–8786.
- (40) Bastin, L.; B́arcia, P. S.; Hurtado, E. J.; Silva, J. A. C.; Rodrigues, A. E.; Chen, B.-L. *J. Phys. Chem. C* **2008**, *112*, 1575–1581.
- (41) Munusamy, K.; Sethia, G.; Patil, D. V.; Rallapalli, P. B. S.; Somani, R. S.; Bajaj, H. C. *Chem. Eng. J.* **2012**, *195–196*, 359–368.
- (42) Kim, S.-N.; Kim, J.; Kim, H.-Y.; Cho, H.-Y.; Ahn, W.-S. *Catal. Today* **2013**, *204*, 85–93.
- (43) Yang, Q.; Vaesen, S.; Vishnuvarthan, M.; Ragon, F.; Serre, C.; Vimont, A.; Daturi, M.; De Weireld, G.; Maurin, G. *J. Mater. Chem.* **2012**, *22*, 10210–10220.
- (44) Huang, H.; Zhang, W.; Liu, D.; Zhong, C. *Ind. Eng. Chem. Res.* **2012**, *51*, 10031–10038.